

Broad Spectral Tuning of Ultra-Low Loss Polaritons in a van der Waals Crystal by Intercalation

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ABSTRACT

Phonon polaritons (PhPs) – light coupled to lattice vibrations – in polar van der Waals (vdW) crystals are promising candidates for controlling the flow of energy at the nanoscale due to their strong field confinement, anisotropic propagation and ultra-long lifetime in the picosecond range¹⁻⁵. However, the lack of tunability of their narrow and material-specific spectral range – the Reststrahlen Band (RB) – severely limits their technological implementation. Here, we demonstrate that intercalation of Na atoms in the vdW semiconductor α -V₂O₅ enables a broad spectral shift of RBs, and that the PhPs excited show ultra-low losses (lifetime of 4 ± 1 ps), similar to PhPs in the non-intercalated crystal (lifetime of 6 ± 1 ps). We expect our intercalation method to be applicable to other vdW crystals, opening the door for the use of PhPs in broad spectral bands in the mid-infrared domain.

In recent years, polar van der Waals (vdW) crystals have become excellent platforms to study and manipulate light at the nanoscale⁶⁻⁹, the goal of the burgeoning field of nano-optics. Their reduced dimensionality and intrinsic anisotropy have allowed the discovery of infrared-active phonon polaritons (PhPs) – light coupled to lattice vibrations – with extraordinary properties. Prominent examples are PhPs in the dielectric hexagonal boron nitride (h-BN) — which uncover ultra-strong confinement, slow propagation, and out-of-plane hyperbolic behavior, allowing the observation of exotic optical phenomena such as ray propagation or hyper-lensing effects¹⁰ —, and in the polar vdW semiconductor α -

MoO₃ — revealing in-plane anisotropic propagation (hyperbolic or elliptic) with ultra-low losses² (lifetimes of 8 ps) —, offering opportunities for planar nanotechnologies aiming a directional control of light-matter interactions at the nanoscale.

However, despite the extraordinary properties of these PhPs in vdW crystals, there is still an important technological drawback for their implementation into nanophotonics technologies, namely the narrow and material-dependent spectral band where they exist (the so-called Reststrahlen band (RB), defined between the transverse (TO) and longitudinal optic (LO) phonon frequencies). Although attempts to spectrally tune the RB in bulk polar materials have been recently tackled^{11,12}, a broad spectral tuning of a RB in a polar vdW material, and consequently of the PhPs supported within it, has remained elusive.

In this work, we introduce intercalation as an efficient route to spectrally tune PhPs in a polaritonic vdW crystal, the metal oxide α -V₂O₅. By near-field imaging of Na-intercalated α -V₂O₅ (forming the crystal α' -(Na)V₂O₅) we reveal a RB shift of ~ 30 cm⁻¹ (60% of the initial RB width), and consequently of the PhPs supported within it. Importantly, PhPs in the intercalated α' -(Na)V₂O₅ crystal exhibit ultra-low loss (lifetimes of 4 ± 1 ps) and in-plane anisotropic propagation, similar to PhPs in pristine α -V₂O₅ (lifetimes of 6 ± 1 ps), evidencing that the intercalation of atoms in between the vdW layers does not significantly affect the polaritonic properties of the crystal.

The optical image in Fig. 1a shows the characteristic rectangular shape of α -V₂O₅ thin flakes studied along this work. This asymmetric geometry is inherited from the strong anisotropy of its crystal lattice^{13,14}, which is illustrated in the diagram of Fig. 1b. As can be seen, the crystalline unit cell of α -V₂O₅ shows an orthorhombic structure where the three inequivalent oxygen positions (denoted O₁₋₃) with respect to the vanadium atom give rise to asymmetric V–O bonds along the three different crystalline axes. This strong biaxial anisotropy is also translated into the optical properties of the crystal (Supplementary Information) as observed in Fig. 1c, where the dielectric permittivity of α -V₂O₅¹⁵ along the three crystalline axes is shown. We can see up to three spectral bands in the plotted range (in color) where at least one of the permittivity components is negative, thus indicating the existence of three RBs (denoted RB₁₋₃) and the possibility of supporting PhPs within them. Given that only one (as in RB₁ and RB₃) or two (as in RB₂) components of the permittivity are negative, instead of the three components (as in isotropic media), PhPs in α -V₂O₅ are expected to be strongly anisotropic with either elliptical or hyperbolic propagation^{2,5,16,17}.

To probe the excitation of PhPs in α -V₂O₅, we performed polariton interferometry^{18,19} using scattering-type scanning near-field optical microscopy (s-SNOM), which yields nanoscale resolved near-field images together with sample topography (Methods). In Figs. 2a-c we show near-field amplitude images of an α -V₂O₅ flake with thickness $d = 105$ nm, taken at frequencies belonging to RB₁ of α -V₂O₅: $\omega_0 = 1031$ cm⁻¹, $\omega_0 = 1026$ cm⁻¹ and $\omega_0 = 1020$ cm⁻¹, respectively. In the images we observe bright fringes parallel to all flake edges, but with different periodicities depending on the crystal direction, indicating

PhPs with in-plane (along the flake) anisotropic propagation^{2,5}. This can be better examined by analyzing the profiles shown on the right of the s-SNOM images in Fig. 2b, where the polariton wavelength λ_p is about 915 nm along the [100] crystal direction and about 800 nm along the [001] direction (note that in polariton interferometry^{18,19} the polaritons excited by the tip propagate away and are back-reflected at the flake edges, giving rise to interference fringes with a spacing $\lambda_p/2$), clearly corroborating an in-plane anisotropic propagation of PhPs. Apart from this extraordinary behavior, the PhPs wavelengths along the [100] and [001] directions also reveal a deep subwavelength-scale polariton confinement $\lambda_p \ll \lambda_0 = 9.75 \mu\text{m}$ (where λ_0 is the infrared illuminating wavelength), which is a key characteristic for their potential use in nanophotonics.

The in-plane anisotropic propagation of PhPs in $\alpha\text{-V}_2\text{O}_5$ can be more clearly observed by plotting the dispersions $\omega(k_i)$ ($i = x, y$, corresponding to [100] and [001] crystal directions) extracted from s-SNOM images taken at single incident frequencies (Fig. 2d). Although they follow similar slopes, there is a clear separation between them verifying that PhPs in RB_1 propagate with in-plane anisotropy. The PhPs phase velocity, $v_{p,i} = \omega_0/k_i$, can be also extracted by plotting the complex-valued wavevector (Supplementary Information), being negative for both directions. Also, the slopes of the dispersion curves (Supplementary Information) give us information on their group velocities ($v_{g,i} = \partial k_i / \partial \omega$)⁻¹, which result to be very small (about $0.0009c$) in both directions (at $\omega_0 = 1026 \text{ cm}^{-1}$, see Sup. Info), showing potential for applications involving light-matter interactions²⁰.

To better describe the properties of PhPs in $\alpha\text{-V}_2\text{O}_5$, we performed nanoscale Fourier transform infrared spectroscopy^{2,21} (nanoFTIR) measurements (Methods) along the [100] and [001] in-plane crystal directions (Fig. 3a, left and right panels, respectively). We observe three spectral bands (RB_{1-3}) exhibiting a series of signal maxima within band limits corresponding to LO and TO phonon frequencies of $\alpha\text{-V}_2\text{O}_5$ (RB_{1-3} in Fig. 1c, Supplementary Information) unveiling the existence of PhPs. In RB_1 we find that *i*) the signal maxima show a different spacing (corresponding to λ_p) along both [100] and [001] directions, and that *ii*) λ_p increases with the frequency. These observations further reveal the in-plane anisotropic (elliptic) propagation and negative phase velocity of PhPs in the RB_1 of $\alpha\text{-V}_2\text{O}_5$. In contrast, in RB_2 we find that *i*) the signal maxima (dashed line in the figure) are only present along the [100] direction (there are no visible fringes along the [001] direction) and that *ii*) λ_p decreases with the frequency. These observations indicate the excitation of PhPs with in-plane hyperbolic propagation and positive v_p . Finally, in RB_3 we find fringes, indicated by signal maxima (dashed lines in the figure), only along the [001] crystal direction (there are no visible fringes along the [100] direction), which points out to a similar in-plane hyperbolic behavior to that in RB_2 but along the orthogonal direction. Overall, these nanoFTIR results confirm that $\alpha\text{-V}_2\text{O}_5$ supports PhPs with in-plane anisotropic propagation (elliptic and hyperbolic along both orthogonal directions), thus adding another member to the library of vdW materials supporting PhPs^{2,4,5}.

As a metal oxide, α -V₂O₅, offers the possibility of being efficiently intercalated with alkali ions²², which entails attractive prospects for energy technologies²³. Intercalation has been recently demonstrated to be a promising low-temperature synthesis strategy to tune the physical and chemical properties of vdW materials such as MoS₂²⁴, Bi₂Se₃²⁵, or black phosphorus²⁶. However, the effect of intercalation on the polaritonic response of a layered crystal has been barely studied, being a switching capability (originated by inducing a minimal polariton propagation in the intercalated area) together with a RB shift of a few wavenumbers the only polaritonic effects reported so far⁴. In the following, we study the effects of intercalation (using Na atoms by single crystal growth (Methods), as a difference to previous works based on aqueous solutions⁴) on the polaritonic response of α -V₂O₅. Figure 3b shows nanoFTIR line scans along the [100] and [001] in-plane directions (left and right panels, respectively) of an intercalated α' -(Na)V₂O₅ flake. Interestingly, we observe a spectral band, RB'₁, that appears clearly red-shifted with respect to RB₁ in α -V₂O₅, and that shows periodic fringes of signal maxima indicating the existence of PhPs. These periodic signal maxima show a different spacing along the [100] and [001] directions, which increases with the frequency, revealing PhPs with in-plane anisotropic propagation (elliptic) and negative phase velocity in α' -(Na)V₂O₅. They are thus similar to PhPs in RB₁ of pristine α -V₂O₅ but shifted to smaller frequencies. To better analyze this effect we plot in Figure 4a the dispersions extracted from several monochromatic s-SNOM images for both pristine and intercalated flakes (Methods). We clearly observe that the dispersion of PhPs in RB'₁ (represented by dark green lines) of intercalated α' -(Na)V₂O₅ is strongly red-shifted (about 30 cm⁻¹ from center to center of both RBs), in comparison to the dispersion of PhPs in RB₁ of α -V₂O₅ (represented by bright green lines). This finding unambiguously demonstrates that a broad spectral shift of a RB can be originated by the intercalation of Na atoms into α -V₂O₅, thus demonstrating a key property for PhPs: spectral tunability via intercalation of the host material.

We note that the nanoFTIR image in Fig. 3b does not show any indication of other RBs that could be associated to RB₂ or RB₃ of α -V₂O₅. To better understand the polaritonic effects induced by the intercalation of Na atoms into α -V₂O₅ and the spectral shift obtained, we calculate from first principles (Supplementary Information) the phonon dispersions of α -V₂O₅, and α' -(Na)V₂O₅ crystals. The crystalline structures used in the calculations are illustrated in the top row of Figure 3, where the typical orthorhombic structure of α -V₂O₅ (left) is modified by the intercalation of Na atoms preferentially located in between the vdW layers (right). The extracted phonon modes (Supplementary Information) are then used in a Lorentz oscillators model to retrieve the theoretical permittivities for both crystalline structures, which are plotted in Figure 4b. In the plotted range, α -V₂O₅ shows three RBs (RB₁₋₃ in the figure) in qualitative agreement with the experimental permittivity shown in Fig. 1c, thus validating our *ab initio* calculations. For α' -(Na)V₂O₅ we also obtain three RBs (RB'₁₋₃), yet narrower and centered at shifted frequencies with respect to RB₁₋₃, respectively. Particularly, RB'₁ appears red-shifted for about 50 cm⁻¹ (from center to center) in good agreement with our results shown in the nanoFTIR images in Fig. 3b. On the other hand, mainly due to a modification of the

oxygen effective charges along the [100] direction, RB'_2 appears blue-shifted and strongly narrower to the point of being almost imperceptible, which explains its absence in our experiments. Finally, RB'_3 is also narrower, being the LO phonon frequency strongly red-shifted, which also explains its absence in our measurements shown in Fig. 3b, as it lays out of our nanoFTIR spectral range.

Apart from the spectral shifts induced in the RBs (and thus in the PhPs dispersions) discussed above, the intercalation of Na atoms into the α - V_2O_5 crystal lattice might have an influence on the PhPs anisotropic propagation and, more importantly, on their lifetimes. To compare the propagation of PhPs in α - V_2O_5 and α' -(Na) V_2O_5 , we fabricated a metal antenna (gold disk) on top of both crystals and imaged their polaritonic activity by s-SNOM at frequencies within RB_1 and RB'_1 , respectively (Figs. 5a, b). Due to its circular geometry, the gold disk can act as an efficient launcher of PhPs along all directions in the plane, thus enabling a direct investigation of the PhPs propagation. In both crystals, we observe fringes along all directions in the plane corresponding to the excitation of PhPs²⁷. Interestingly, they both show a slightly longer λ_p along the [001] direction, revealing similar in-plane elliptic propagation of PhPs in both crystals. From this, we can draw the conclusion that intercalation has a negligible effect on the in-plane anisotropic propagation of PhPs in α - V_2O_5 .

Finally, we study and compare the PhPs lifetimes in both pristine α - V_2O_5 and intercalated α' -(Na) V_2O_5 crystals, which is crucial for validating intercalation as an effective material synthesis strategy for nanophotonics. To do this, we extracted s-SNOM line profiles (red crosses in Figs. 5c, d, taken at $\omega_0 = 1010 \text{ cm}^{-1}$ and $\omega_0 = 973 \text{ cm}^{-1}$, respectively) along the [001] direction of the α - V_2O_5 , and α' -(Na) V_2O_5 flakes with similar thicknesses of 130 and 107 nm, shown in Figs. 5a, and b, respectively. By fitting them with an exponentially decaying sine-wave function corrected by the geometrical spreading factor^{2,28} \sqrt{x} (Supplementary Information) we extract decay lengths $L_{p[001]}$ of $1.40 \mu\text{m}$ for α - V_2O_5 PhPs (yielding a propagation figure of merit³ $Q = \text{Re}(k)/\text{Im}(k)$ of 3.5) and $1.15 \mu\text{m}$ for α' -(Na) V_2O_5 PhPs (yielding $Q = 2.5$). With these values we obtain the lifetimes according to $\tau_{[001]} = L_{p[001]} / v_g$, where the group velocities v_g are taken from the PhPs dispersions (Fig. S4 in the Sup. Info.). We obtain $\tau_{[001]} = 6 \pm 1 \text{ ps}$ for PhPs in α - V_2O_5 , and $\tau_{[001]} = 4 \pm 1 \text{ ps}$ for PhPs in intercalated α' -(Na) V_2O_5 . These notable values of the lifetimes (obtained despite a short L_p due to the ultra-slow PhPs group velocity of $0.0007c$) reveal the low-loss nature of PhPs in α - V_2O_5 , which can be related to the fact that α - V_2O_5 is close to be isotopically pure, since natural abundance oxygen and vanadium have both isotopic purities of 99.7%, and, more importantly, that the intercalation process followed in this work not only allows for a large spectral shift of PhPs, but also for preserving their low-loss nature (despite a significant reduction, the lifetime is still in the range of ps).

In conclusion, this work demonstrates intercalation of atoms (Na) in a vdW crystal (α - V_2O_5) as an efficient technological approach to achieve a broad spectral shift of PhPs with ultra-long lifetimes. Considering that a large variety of ions and ion contents can be

intercalated in layered materials²⁹, we envision on-demand spectral response of PhPs in vdW materials, eventually allowing for covering the whole mid-IR range, critical for the emerging field of phonon polariton photonics. We also note that α -V₂O₅, as a semiconductor, can be electrically doped to support plasmon polaritons, which, via coupling to LO phonons³⁰, could eventually allow for a dynamic tuning of PhPs spectrally shifted by intercalation.

DATA AVAILABILITY

The data that support the findings of this study are available from the corresponding authors on reasonable request.

CODE AVAILABILITY

The code employed in this work to perform all calculations is available from the corresponding authors on reasonable request.

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METHODS

s-SNOM

We used a scattering type scanning near-field optical microscope (s-SNOM, from Neaspec GmbH) for infrared nanoimaging. This system employs a Pt-Ir coated atomic force microscope (AFM) tip illuminated with infrared light as a launcher and recorder of polaritons, yielding simultaneously near-field images and topography. The metallized AFM tip is illuminated with light (p-polarized) of frequency ω from a quantum cascade laser. The tip oscillates at a frequency $\Omega \approx 270$ kHz with an amplitude of about 100 nm. The tip-launched polaritons reflect at the V_2O_5 flake edges and produce polariton standing wave interferences, which are imaged by recording the light scattered by the tip. This is carried out with a pseudo-heterodyne interferometer that demodulates the detector signal at high harmonics $n\Omega$ (typically 3rd harmonics named as s_3) and provides background-free detection. The metal coated AFM tip provides wavelength independent resolution.

nanoFTIR

For the nanoFTIR images, we employed the nanoFTIR module for the s-SNOM system. We used an Au coated AFM tip illuminated by a super-continuum laser and the scattered light was recorded with an asymmetric Fourier transform spectrometer. Spectral line scans are obtained by recording point spectra as a function of the tip position.

α - V_2O_5 and α' -(Na) V_2O_5 sample growth and preparation

Orthorhombic layered α - V_2O_5 with Van der Waals structure was synthesized via a purification method and subsequent single crystal growth, as described in references [14, and 31]. Sodium-intercalated α' -(Na) V_2O_5 single crystals were grown in a similar way as described in reference [32]. Briefly, a mixture of $NaVO_3$ (1.843g) and VO_2 (0.1564g) powder was put into a Pt crucible with a lid and sealed under vacuum in a quartz tube. The mixture was heated to 800 °C at a rate of 100 °C/h and kept at 800 °C for 0.5 h. It

was then cooled to 740 °C at a rate of 1 °C/h, kept at 740 °C for 3 days, slowly cooled to 600 °C at a rate of 1.5 °C/h, and finally cooled to room temperature at a rate of 300 °C/h to yield α' -(Na)V₂O₅ crystals. Temperature dependent magnetic susceptibility measurement of the as-grown α' -(Na)V₂O₅ crystal verified a spin-Peierls transition at ~35K, in agreement with the literature³² and indicating adequately intercalated α' -(Na)V₂O₅. Bulk α -V₂O₅ and α' -(Na)V₂O₅ crystals were thinned down by mechanical exfoliation using Nitto blue tape leading to thin slabs with thicknesses in the range from 100 nm to a few micrometers, which are transferred on top of a 300-nm-thick SiO₂ layer deposited on a Si substrate. For maximum yield of thin devices, substrates were heated to 90 °C for 10 minutes during transfer.

Dielectric function of α -V₂O₅

The principal values of the diagonal permittivity tensor, ϵ_x , ϵ_y and ϵ_z are approximated with a three-parameter Drude–Lorentz permittivity:

$$\epsilon_a(\omega) = \epsilon_{a,\infty} \frac{(\omega_{LO}^a)^2 - \omega^2 + i\gamma^a \omega_{LO}^a \omega}{(\omega_{TO}^a)^2 - \omega^2 + i\gamma^a \omega_{TO}^a \omega}$$

where $a = x, y, z$, ω_{TO} and ω_{LO} refer to the transversal (TO) and longitudinal (LO) phonon frequencies, respectively, γ denotes the damping constant and $\epsilon_{a,\infty}$ is the high frequency permittivity. The values of the constants are: $\omega_{TO}^x = 765 \text{ cm}^{-1}$, $\omega_{LO}^x = 952 \text{ cm}^{-1}$, $\gamma^x = 40 \text{ cm}^{-1}$, $\epsilon_{x\infty} = 6.6$; $\omega_{TO}^y = 506 \text{ cm}^{-1}$, $\omega_{LO}^y = 842 \text{ cm}^{-1}$, $\gamma^y = 19 \text{ cm}^{-1}$, $\epsilon_{y\infty} = 6.1$; $\omega_{TO}^z = 976 \text{ cm}^{-1}$, $\omega_{LO}^z = 1037 \text{ cm}^{-1}$, $\gamma^z = 2.0/1.5 \text{ cm}^{-1}$, $\epsilon_{z\infty} = 3.9$.

The values of ω_{LO}^a and ω_{TO}^a were adjusted by fitting FTIR and s-SNOM measurements with transfer-matrix calculations, while $\epsilon_{a,\infty}$ were obtained from *ab initio* calculations (Supplementary Information). The values of γ^a were taken from ref. [15]. $\gamma^z = 1.5 \text{ cm}^{-1}$ is also considered (Supplementary Information) to better fit the experimental lifetime values and in analogy to ref [33].

Structural Characterization

α -V₂O₅ and α' -(Na)V₂O₅ local structures were analyzed by High Resolution Transmission Electron Microscopy (HRTEM) using a Tecnai F20 ST (FEI) operating at 200 kV. X-ray diffraction was performed using a Bruker Discover D8 with VANTEC-500 solid state detector, using a Co K-alpha X-ray source with wavelength 1.788 nm.

Methods References

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ACKNOWLEDGEMENTS

J.T-G and G.Á-P acknowledge support through the Severo Ochoa Program from the Government of the Principality of Asturias (PA-18-PF-BP17-126 and PA-20-PF-BP19-053, respectively). J.M-S. acknowledges support through the Clarín Programme from the Government of the Principality of Asturias and a Marie Curie-COFUND grant (PA-18-ACB17-29). K.C., X.P.A.G., H.V., and M.H.B. acknowledge the Air Force Office of Scientific Research (AFOSR) Grant FA 9550-18-1-0030 for funding support. I.E. acknowledges financial support from the Spanish Ministry of Economy and Competitiveness (FIS2016-76617-P). A.Y.N. acknowledges the Spanish Ministry of Science, Innovation and Universities (national project MAT2017-88358-C3-3-R) and Basque Government (grant No. IT1164-19). Q. B. acknowledges the support from Australian Research Council (ARC, FT150100450, IH150100006 and CE170100039). P.A-G. acknowledges support from the European Research Council under Starting Grant 715496, 2DNANOPTICA.

AUTHOR CONTRIBUTIONS

P.A-G. and J.T-G. conceived the study. P.A-G. and J.M-S. supervised the project. J.T-G, and J.D carried out the near-field imaging experiments with the help of M.A., S.L. and W.M. G.A-P., A.B., R.H., Q.B. and A.Y.N. participated in data analysis. P. A-G wrote the manuscript with input from J.T-G, G.A-P, J.M-S, A.Y.N, J.D, Q.B, I.E., X.P.A.G, K.C. and R.H. I.E carried out *ab initio* calculations. G.A-P. and A.Y.N. conducted the analytical calculations. K.K., T.K., K.C., and X.P.A.G. contributed to material synthesis and sample preparation. H.V. and M.H. B. performed the TEM characterization. K.C. performed XRD indexing and characterization. I.P. contributed to sample fabrication.

COMPETING FINANCIAL INTERESTS

R.H. is co-founder of Neaspec GmbH, a company producing scattering-type scanning near-field optical microscope systems, such as the one used in this study. The remaining authors declare no competing interests.

FIGURES

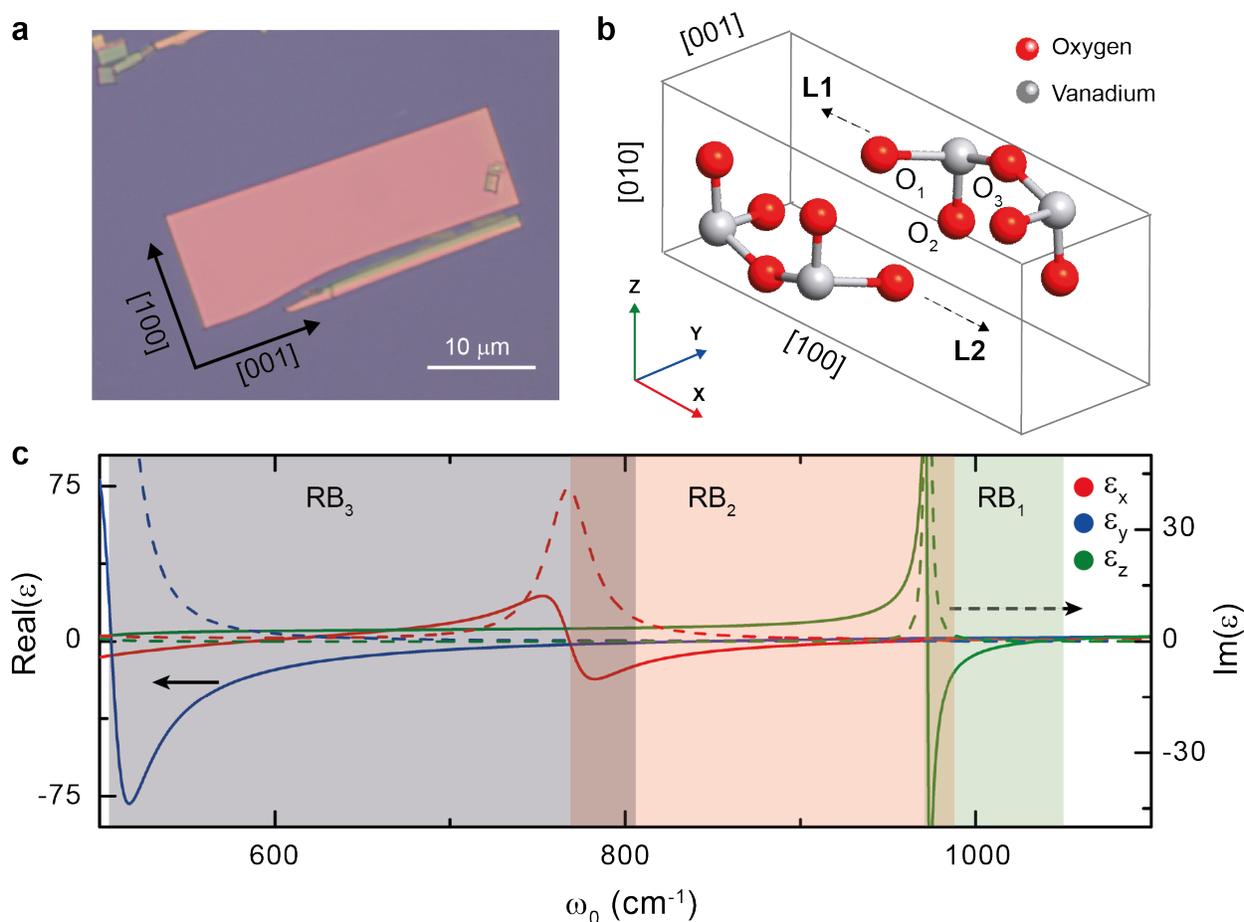


Figure 1: Physical properties of α - V_2O_5 . **a.** Optical image of an α - V_2O_5 flake. α - V_2O_5 crystals show a rectangular shape owing to their anisotropic crystal structure. Labelled arrows indicate the [100] and [001] crystal directions. **b.** Schematic of the crystalline unit cell of α - V_2O_5 ; the lattice constants along the principal x, y, and z axes are $a = 1.15$ nm, $b = 0.36$ nm and $c = 0.44$ nm, respectively. Grey spheres represent the vanadium atoms and red spheres represent the oxygen atoms. The three oxygen positions of asymmetric V–O bonds along different crystalline axes are denoted O_{1-3} . L1 and L2 indicate the two vdW layers weakly bound along the [010] crystal direction. **c.** Real (continuous lines) and imaginary part (dashed lines) of the permittivity (see Methods) along the principal x ([100]), y ([001]), and z ([010]) axes (red, blue, and green lines, respectively). The Reststrahlen bands RB_1 , RB_2 , and RB_3 are indicated in green, red and grey, respectively.

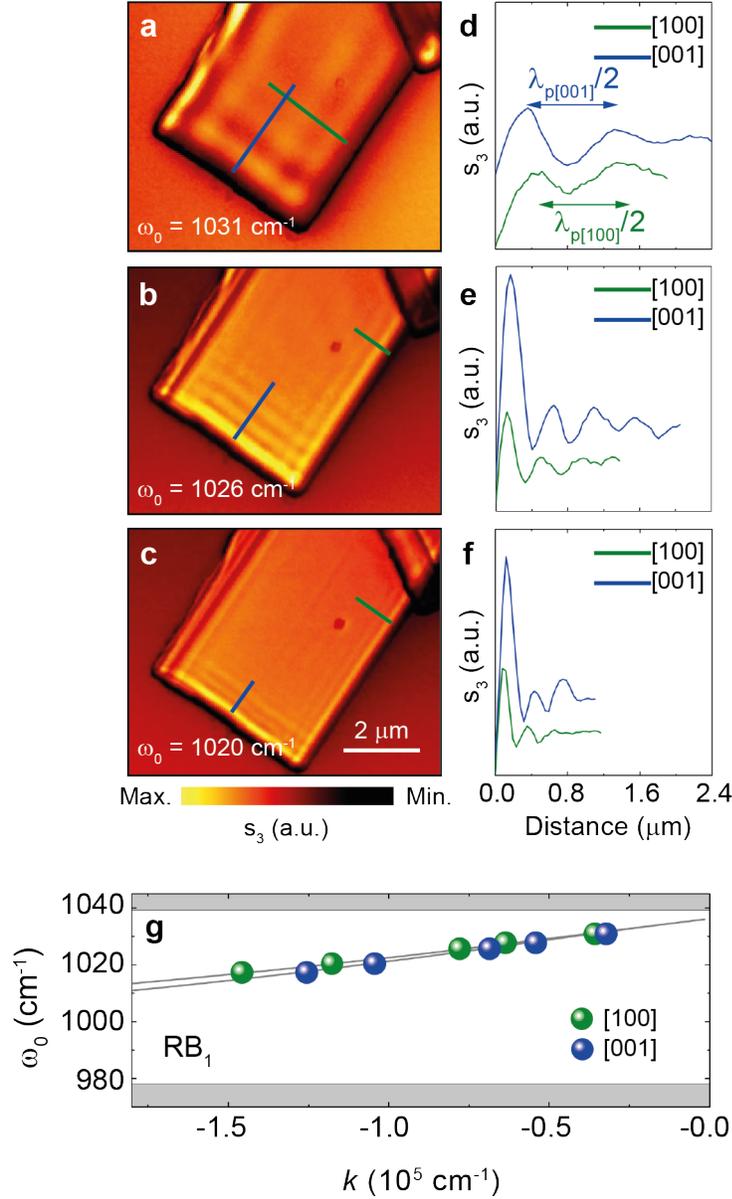


Figure 2: Real-space imaging of a α - V_2O_5 flake. **a-c.** Near-field amplitude images s_3 of an α - V_2O_5 flake with thickness $d = 105 \text{ nm}$ at incident frequencies $\omega_0 = 1031 \text{ cm}^{-1}$, 1026 cm^{-1} , and 1020 cm^{-1} , respectively. **d-f** Profiles along the [100] (green lines) and [001] (blue lines) directions, extracted from the near-field amplitude images in **a-c**, respectively. $\lambda_{p[100]}$ and $\lambda_{p[001]}$ are the polariton wavelength along the [100] and [001] direction, respectively. **g.** Dispersion of PhPs along the [100] (green symbols) and [001] (blue symbols) directions in the RB_1 . Grey lines are guides to the eye. Grey shaded areas indicate the spectral regions outside the RB .

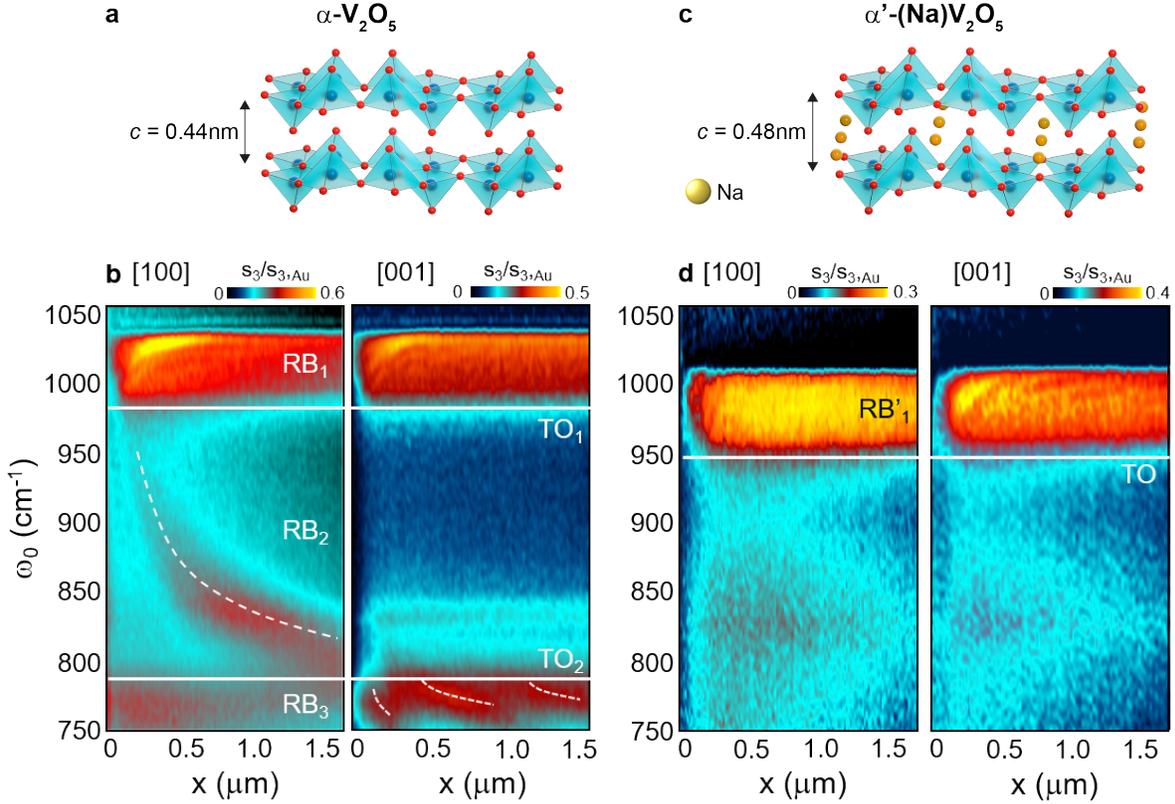


Figure 3: Real-space nano-spectroscopy of α - V_2O_5 and intercalated α' - $(Na)V_2O_5$ flakes. **a.** Illustration of the α - V_2O_5 lattice structure (orthorhombic) where the red spheres represent oxygen atoms, the blue atoms represent vanadium atoms, and the blue pyramids show the polyhedral structure defined by the oxygen atoms. The crystal structure consists of bilayers of distorted VO_5 pyramids stacked along the [010] direction via vdW interactions (interlayer distance $c = 0.44$ nm). **b.** nano-FTIR spectral line scans along the [100] and [001] directions of a α - V_2O_5 flake showing $s_3/s_{3,Au}$ (near-field amplitude s_3 normalized on Au, $s_{3,Au}$) as a function of distance between the tip and the flake edge. Solid horizontal lines mark the approximate transversal phonon modes in α - V_2O_5 (TO_1 , 975 cm^{-1} ; TO_2 , 770 cm^{-1}), separating RB_{1-3} . Dashed lines are guides-to-the-eye of signal maxima. The flake thickness is $d = 245$ nm. **c.** Illustration of the α' - $(Na)V_2O_5$ lattice structure (orthorhombic) where the red spheres represent oxygen atoms, the blue atoms represent vanadium atoms, the yellow atoms represent sodium atoms, and the blue pyramids show the polyhedral structure defined by the oxygen atoms. The crystal structure consists of bilayers of distorted VO_5 pyramids with sodium atoms intercalated and stacked along the [010] direction via vdW interactions (interlayer distance $c = 0.48$ nm). **d.** nano-FTIR spectral line scans along the [100] and [001] directions of a α' - $(Na)V_2O_5$ flake showing $s_3/s_{3,Au}$ (near-field amplitude s_3 normalized on Au, $s_{3,Au}$) as a function of distance between the tip and the flake edge. Solid horizontal lines approximately mark the transversal phonon modes in α' - $(Na)V_2O_5$ (TO , 950 cm^{-1}), defining RB'_1 . The flake thickness is $d = 150$ nm. The scale in the colour bars of **b** and **d** is linear.

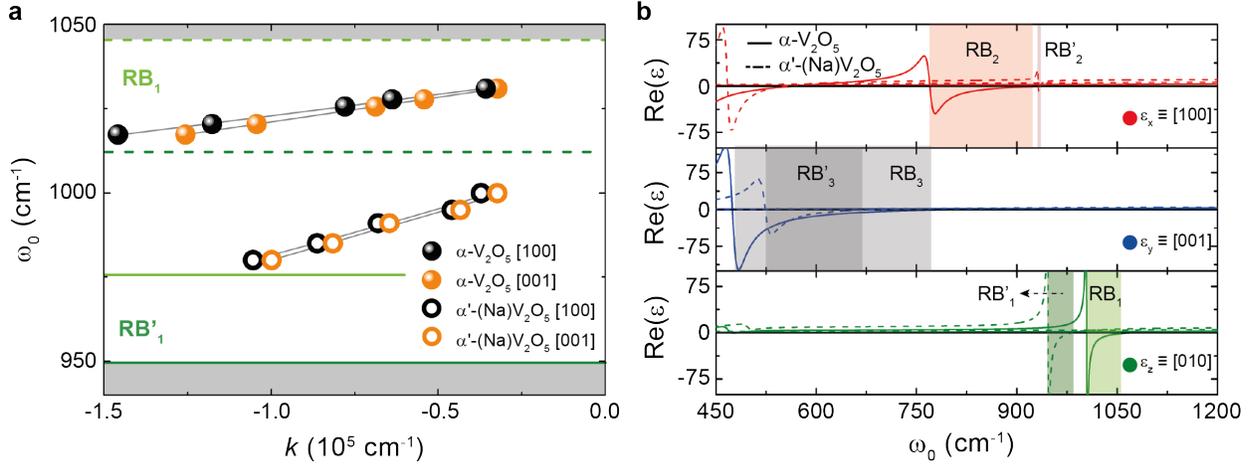


Figure 4: PhPs dispersion and *ab initio* permittivity in α - V_2O_5 and intercalated α' -(Na) V_2O_5 crystals. **a.** Dispersion of PhPs along the [100] and [001] directions in a α - V_2O_5 (full symbols) flake with thickness $d = 105$ nm and a α' -(Na) V_2O_5 (empty symbols) flake with thickness $d = 190$ nm. Dashed and continuous horizontal lines mark the approximate TO and LO phonon modes in α - V_2O_5 (TO, 980 cm^{-1} ; LO, 1040 cm^{-1}), and α' -(Na) V_2O_5 (TO, 945 cm^{-1} ; LO, 1015 cm^{-1}), respectively. Grey lines are guides to the eye. Grey shaded areas indicate the spectral regions outside the RBs. **b.** Real-part of the permittivities for α - V_2O_5 (continuous lines) and α' -(Na) V_2O_5 (dashed lines) extracted from *ab initio* calculations along the principal x, y, and z axes (red, blue, and green lines, respectively). The Reststrahlen bands RB_{1-3} , and RB'_{1-3} for α - V_2O_5 and α' -(Na) V_2O_5 , are indicated in bright and dark color, respectively. Green shaded regions represent RB_1 and RB'_1 ; Red shaded regions represent RB_2 and RB'_2 and grey shaded regions represent RB_3 and RB'_3 .

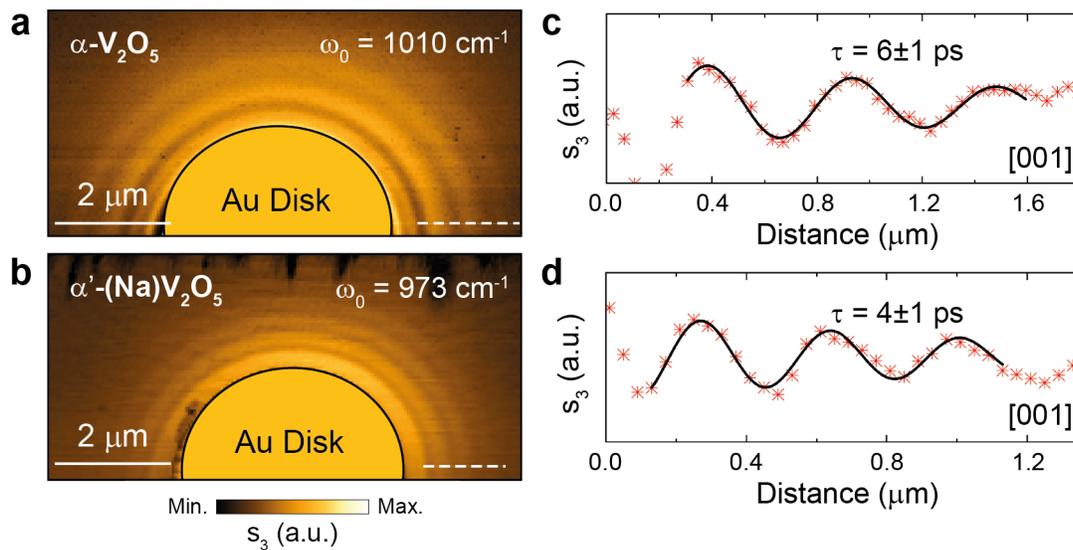


Figure 5: Anisotropy and lifetimes of PhPs in α - V_2O_5 and intercalated α' -(Na) V_2O_5 flakes. **a,b.** Near-field amplitude images s_3 of α - V_2O_5 and α' -(Na) V_2O_5 flakes with thicknesses $d = 130$ nm, and $d = 107$ nm at illuminating frequencies $\omega_0 = 1010$ cm^{-1} (RB_1) and $\omega_0 = 973$ cm^{-1} (RB'_1), respectively. A gold disk (half of it shown in the image for convenience) is used as an antenna for efficient launching of PhPs along all in-plane directions. **c,d.** s-SNOM line traces (showing the amplitude s-SNOM signal σ_3 ; Methods) along the [001] direction of α - V_2O_5 and α' -(Na) V_2O_5 flakes in a and b (white dashed lines). Damped sine-wave functions (black solid lines) were fitted to the data (Supplementary Information). Lifetimes $\tau = 6 \pm 1$ ps and $\tau = 4 \pm 1$ ps are obtained, respectively.